

# Surface properties of InN and InGaN alloys

P. D. C. King, T. D. Veal, I. Mahboob, L. F. J. Piper, M. Walker and C. F. McConville<sup>†</sup>

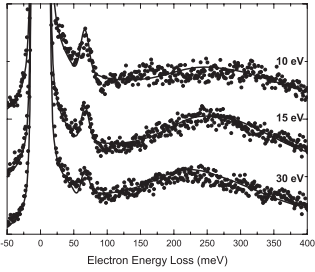
Department of Physics, University of Warwick, Coventry, CV4 7AL, United Kingdom

## Introduction

Indium nitride (InN) remains one of the least understood of the III-V semiconductor systems, despite its enormous potential for device applications. However, InN and In-rich III-N alloys are known to have a number of striking fundamental properties, particularly at the free surface. These surface properties are investigated here.

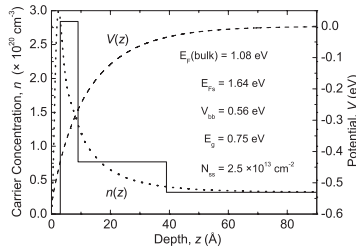
## Electron accumulation at InN surfaces

### Intrinsic electron accumulation at InN(0001) surface



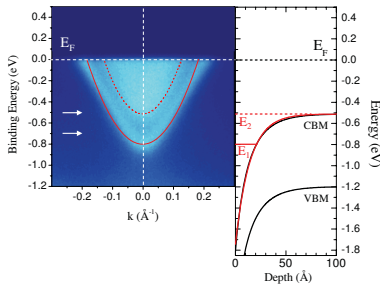
- Realistic charge and potential profile determined by solving Poisson's equation within the MTFA including non-parabolicity

- HREEL spectra: Plasmon peak dispersion → surface electron accumulation
- Layered profile used to simulate HREELS spectra



I. Mahboob *et al.*, Phys. Rev. Lett. **92**, 036804 (2004).

### Quantized electron accumulation



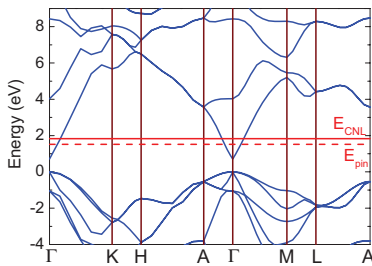
- Downward band bending causes 1D quantum well (approximately exponential) near surface
- CB states become quantized at surface (quasi 2DEG)
- Quantization direction is perpendicular to surface
- Subbands can be resolved by ARPES

L. Colakerol *et al.*, Phys. Rev. Lett. **97**, 237601 (2006).

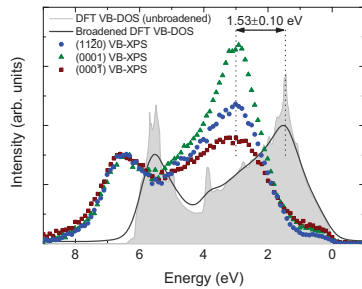
### Origin of the electron accumulation

- Charge neutrality level (CNL) marks cross-over from donor-like (below) to acceptor-like (above) surface states
- Low  $\Gamma$ -point CBM → CNL high above CBM
- $E_F$  at surface below CNL → unoccupied donor ViGS and positive surface charge
- Surface  $E_F$  pinned close to CNL: bands bend downwards, space-charge balances surface charge, extreme electron accumulation

I. Mahboob *et al.*, Phys. Rev. B **69**, 201307 (2004); P. D. C. King *et al.*, Phys. Rev. B (in press).



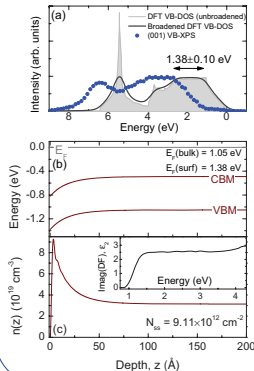
### Universal electron accumulation



- Compare VB-XPS with quasi-particle corrected DFT VB-DOS [1]
- VBM to surface  $E_F$  is  $1.53 \pm 0.10$  eV for In- and N-polar *c*-plane and *a*-plane InN
- Surface  $E_F$  pinned in same place for different surface orientations
- $N_{ss} \approx 1.65 \times 10^{13} \text{ cm}^{-2}$
- Differences in intensity due to surface DOS [2]

P. D. C. King *et al.*, Appl. Phys. Lett. **91**, 092101 (2007).

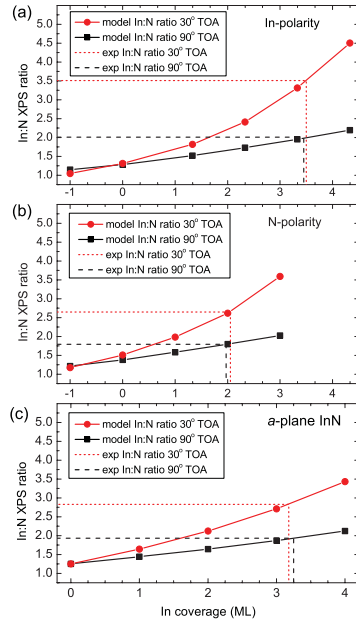
### Zinc-blende InN



- MBE grown *zb* InN with only ~5% *wz* inclusions [3]
- VB photoemission for surface  $E_F$ , ellipsometry (P. Schley & R. Goldhahn, TU Ilmenau) for bulk  $E_F$
- Surface  $E_F >$  bulk  $E_F$  → surface electron accumulation
- zb* surface  $E_F <$  *wz* surface  $E_F$ ; surface state density may be lower for *zb* than for *wz* phase
- $\Gamma$ -point CBM low for *zb*-InN as well as *wz*-InN
- CNL above CBM
- Surface electron accumulation as for *wz*-InN

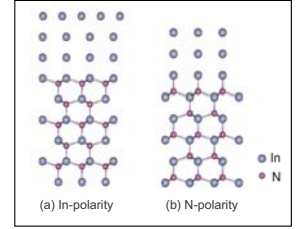
## In-adlayers

Recent first principles calculations predict In-adlayers are the most favourable surface reconstruction for InN [4], similar to GaN under Ga-rich conditions. In-In bonds in an In-adlayer have been suggested as the microscopic origin for the electron accumulation [5].

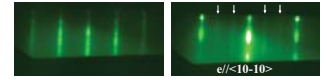


T. D. Veal *et al.*, Phys. Rev. B **76**, 075313 (2007); T. D. Veal *et al.*, Physica B **401-402**, 351 (2007).

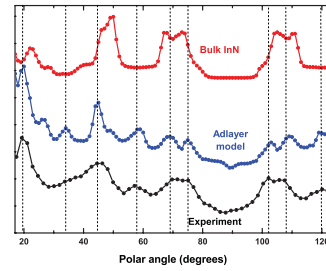
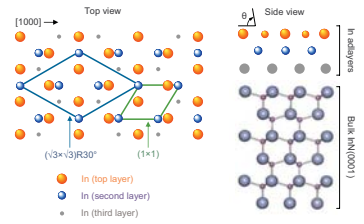
- XPS In:N ratio → ~3.4 ML (~2.0 ML) of In on In- (N-) polarity *c*-plane InN
- Both 1 ML more than for Ga adlayers on *c*-plane GaN under Ga-rich conditions



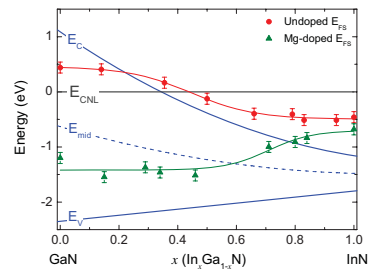
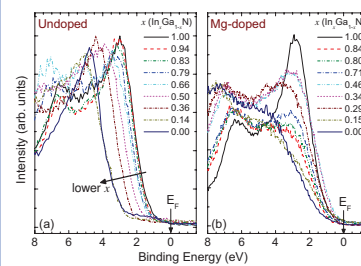
- a*-plane InN has ~3 ML In above non-polar bulk-like termination
- Fractional In-adlayer coverage for InN(0001) implies laterally contracted top layer
- RHED of an InN(0001) sample shows  $1 \times$  and  $3 \times \rightarrow (\sqrt{3} \times \sqrt{3})R30^\circ$  reconstruction



- Co-axial impact collision ion scattering spectroscopy supports this



## InGaN alloys



- For undoped alloys, transition from electron accumulation (In-rich) to depletion (Ga-rich) occurs at  $x \approx 0.43$  with a change in direction of band bending
- For Mg-doped alloys, downward band bending across composition range, but transition from inversion (In-rich) to hole depletion (Ga-rich) occurs at  $x \approx 0.59$
- Form of space charge regions and difference in barrier height of undoped and Mg-doped alloys give evidence for *p*-type bulk conductivity across the composition range

T. D. Veal *et al.*, Appl. Phys. Lett. **89**, 202110 (2006); P. D. C. King *et al.*, Phys. Rev. B **75**, 115312 (2007); P. D. C. King *et al.*, Phys. Stat. Sol. (submitted).

## Conclusions

- Intrinsic electron accumulation at InN surfaces due to low  $\Gamma$ -point CBM
- Extreme downward band bending → quantized CB states at surface
- Core-level XPS and ion scattering reveal In-adlayer termination for *wz*-InN
- Transitions in space-charge type observed with InGaN alloy composition

## References

- F. Fuchs *et al.*, Phys. Rev. B **76**, 115109 (2007)
- T. D. Veal *et al.*, Phys. Rev. B **76**, 075313 (2007)
- J. Schörmann *et al.*, Appl. Phys. Lett. **89**, 261903 (2006)
- D. Segev and C. G. Van de Walle, Surf. Sci. **601**, L15 (2007).
- D. Segev and C. G. Van de Walle, Europhys. Lett. **76**, 305 (2006).

<sup>†</sup>Electronic address: C.F.McConville@warwick.ac.uk